ESR STUDIES ON POLYMERS WITH PARTICULAR ELECTRONIC AND MAGNETIC PROPERTIES

Hans-Klaus Roth¹ and Viktor I. Krinichnyi²

[']Leipzig University of Technology, Department of Natural Sciences, PF 66, D-O-7030 Leipzig, Germany ²Institute of Chemical Physics, Russian Academy of Sciences, 142 432 Chernogolovka, Russia

<u>Abstract:</u> Measurements on insulating and conducting polymers from the polyaniline (PANI) family and investigations on semiconducting poly(tetrathiafulvalenes) (PTTF) have been used for illustration and discussion of some methodical questions of electron spin resonance (ESR). This concerns especially the new possibilities of the recently developed high-resolution ESR in the 2 mm wave band. It was applied for the study of the nature and dynamics of paramagnetic centres (PC) and charge carriers in PANI and PTTF. The rate of the quasi-one-dimensional (ID) intrachain electron motion and of the three-dimensional (3D) interchain electron hopping has been estimated separately. In iodine-doped PTTF the maximum electrical conductivity is 10^{-4} S cm⁻¹. It is almost identical with the 3D-conductivity estimated by ESR and shows the typical temperature dependence of a semiconductor. The quasi-ID-conductivity is several orders of magnitude higher and shows in its dependence on temperature similarities with a metal. The PANI samples show in the highly doped form a maximum conductivity, supporting the model of metallic islands.

1. Introduction

Electron spin resonance (ESR) is a unique spectroscopic method with a broad application field in polymer sciences (Ref.l). It is very sensitive in the study of the behaviour of paramagnetic molecules or molecule segments in polymers. Especially, it allows to get singular information about principal properties of polymers with particular electronic and magnetic properties like conducting polymers and polymers with antiferro-, ferri- and ferromagnetic properties. The study of such systems requires the use of all potentialities of ESR spectroscopy including the electron spin echo (ESE) detection and the high-field (and high-frequency) ESR. This will be shown in connection with our studies on semiconducting poly(tetrathiarulvalencs) and in connection with measurements on insulating and conducting polyanilines.

2. Eiperimental

2.1. The investigated polymers

<u>Polv(tetrathiafulvalene) PTTF</u> is a sulfur-organic polymer which shows long-term stability and a significant increase in electrical conductivity upon doping. In the studied poly(tetrathiafulvalenes) the TTF units are linked via phenyl groups or via tetrahydroanthracene bridges (Ref.2). The PTTF is doped with oxygen and with iodine, respectively. The electrical conductivity strongly depends on the temperature as typical for semiconducting materials. The chemical structure of the polymers investigated is the following:



TABLE 1

The substituents R, the average molecular weights m and the average number of monomeric units π of the investigated compounds

Compound	Structure	R	т	n
A.I	А	-H	7150	25
A.2	А	-CH ₃	7660	25
A.3	А	-CH2CH3	7800	25
B.I	В		6810	20

Details of the synthesis are given in Refs.2,3. All these amorphous polymers are usually obtained in powder form. The samples were doped with iodine dissolved in chloroform using different concentrations of the dopant. The iodine content is referred to the TTF unit (iodine per monomeric unit).

144

<u>Polyaniline (PANI)</u> is one of the most important electrically conducting polymers because of the unique combination of high conductivities, environmental stability and good processibility. The molecular structure and some properties of this polymer depend on the synthesis route. The studied PANI have been synthesized by Lux (Ref.4) via a modification of the general route with (MH^5rO3 in 1.0 M hydrochloric acid as described, e.g., by Cao and Tan(Refs.5,6).

After the synthesis the PANI powder was neutralized in 1.0 M NH4OH. The basic polymer (PANI I in the undoped form (y = 0)) is a poly(phenylenamine) or a leucoemeraldine base polymer (y is the doping level).



The applied doping process is an oxidation. The highly doped PANI (predominantly PANI II at y > 0.1) is a polyaniline with sulfate ions as counter ions (Ref.4). It contains probably as well positively charged radical ions (polarons) as bipolarons.



The aim of the ESR measurements is to obtain more information about the kind and properties of spins and charge carriers in poly(tetrathiafulvalenes) and polyaniline.

2.2. Basic requirements of ESR spectroscopy

A basic requirement for the application of ESR techniques to the polymer characterization is the presence of molecules containing unpaired electron spins. Most conducting polymers contain several types of paramagnetic centres (PCs) which differ in mobility and other properties. Frequently one part of the charge carriers responsible for the electrical conductivity owns a spin, that is they are paramagnetic species. Other unpaired electron spins can be fixed on segments of the polymer chains. Their mobility is determined predominantly by the motions of the chain segments.

Both groups of paramagnetic centres are to be found in the conducting polymers PANI and PTTF. They will be used for a deeper characterization of the conducting polymers included the intrachain electron diffusion, interchain hopping and anisotropic slow segment motions.

2.3. Advantages of high-field ESR demonstrated in connection with measurements on PANI

In the 3 cm wave band the ESR spectra of organic free radicals are registered in a narrow range of the magnetic field where the exchange rate between distinguished spin packets is high in comparison with its width and an overlapping effect exists. As shown in previous papers (Refs.7-10), in the 2 mm wave band the spectral resolution and information content of the ESR method increases considerably and allows more complete investigations of the structure and dynamic properties of paramagnetic centres (PCs) in conjugated polymers. The ESR experiments on PANI have been performed by Krinichnyi et al. (Ref.1 1) using a 3 cm wave band PS-100X and 2 mm wave band ESR-05 (Ref. 12) spectrometers with 100 kHz field *a.c.* modulation for phase lock detection. The total spin concentration in the samples was determined using CuSO₄-5H₂O single crystal as a standard. Another standard, Mn²⁺ with $g_{eff} = 2.00102$ and a = 8.74 mT, was used for the determination of the *g*-factor as well as for the magnetic field sweep scale calibration in the 2 mm wave band.



Fig 1: 3 cm wave band ESR spectra of four PANI samples with various doping levels y measured at room temperature and normalized to equal amplitudes.

The 3 cm ESR spectra are superpositions of two signals with different anisotropy caused by two types of spins with various mobilities. The strong asymmetric signal components seen in the low field region of the ESR spectra of the undoped and very low doped samples ($y \ u \ 0.04$) decreases by oxidation to zero if the y-values become higher than 0.1 and the main ESR signal becomes narrower and is shifted to higher fields if $y \ge 0.2$. This might be due to an increase of mobility of PCs and to bipolaron formation. The averaged g-factor changes from g = 2.0036 to g = 2.0027. This is accompanied by a strong increase of the amplitude of the narrow signal. It reflects an increase of spin carriers with high mobility which fits very well with the increase of electrical conductivity by doping.



Fig. 2: 2 mm wave band ESR spectra of PANI samples with various doping levels measured at room temperature; a) undoped and very lowly doped samples, b) highly doped samples.

Also the 2 mm ESR spectra of the two PANI samples characterized by extremely low conductivity and y < 0.04 are superpositions of two signals: of a weak, very asymmetric signal of strongly localized PCs R_I with the *g*-tensor components $g_x = 2.00535$, $g_y = 2.00415$, $g_z = 2.00238$, and the components of the hyperfine structure tensor $A_x = A_y = 0.33$ mT and $A_z = 2.3$ mT, and of a less asymmetric signal of higher intensity belonging to the more mobile PCs R_{II} - For the last the following magnetic parameters were obtained from the simulation: $g_{\uparrow} = 2.00351$, $g_{\parallel} = 2.00212$ (Ref.II). The relative content of the PC R_I in the undoped sample is approximately 25% at room temperature.

Krinichnyi et al. showed earlier (Refs. 10,13,14) that the diffusion of paramagnetic centres with anisotropic g-factors in conjugated organic polymers leads to the averaging of the principle g-tensor components, if

 $\mathbf{n}_{\mathrm{diff}} >> \mid g_{\wedge} - g_{\parallel} \mid \mathbf{b} B_0 / h$

where n_{diff} is the diffusion rate of paramagnetic centres, **b** is the Bohr magneton. It means, paramagnetic centres R_{II} giving rise to the narrow ESR line of figures 1 and 2b possess a relatively high diffusion rate along the polymer chains and may be attributed to mobile spin charge carriers, namely polarons, while at lowly doped samples (y < 0.1) the predominant part of paramagnetic centres exhibiting asymmetric spectra is relatively stron^y localized on segments of polymer chains.

The 2 mm spectra analysis undoubtedly confirms the existence of localized paramagnetic centres with anisotropic g-factors, especially in PANI samples with y < 0.1, and of mobile ones, diffusing along the chain with a rate of $n_{diff} > 5 \times 10^{10} \text{ s}^{-1}$ (Ref. 11). The observed narrow ESR spectra at high doping level are evidence for the formation of areas with high spin density in which a strong spin-spin exchange occurs.

Fig. 2b shows the 2 mm ESR spectra of the highly doped PANI samples with y = 0.2 and 0.4, which in the 3 cm wave band gives rise to a narrow ESR signal almost identical for the two different samples. It shows best the effect of the higher spectral resolution of the 2 mm technique. The complete evaluation of these spectra and of other ESR measurements on PANI are still in progress and will be published soon in a separate paper (Ref. 11).

3. Studies on PTTF by ESR and ESE in the 3 cm waveband

3.1. Results of ESR studies

Previous ESR studies (Refs. 15-17) have shown that the spectra of weakly doped PTTF are a superposition of a strongly asymmetric spectrum of immobile radicals R^i and of a symmetric spectrum caused by mobile cation radicals R^m (formula will be given further down). The dependence of the signal intensities on temperature and doping level has been interpreted by means of a model based on trapped radicals, mobile polarons and bipolarons (Refs. 16,17). In poly(tetrathiafulvalene) the measured g-factors of free radicals are with 2.0068 < g < 2.0081 larger than in PANI (see chapter 2.3.) and in poly(thiophene) with g = 2.0026 (Ref. 18). The high value of the g-factor in PTTF refers to radicals with a large spin density near sulfur atoms. ESR spectra of two phenylene-bridged PTTF weakly-doped only with oxygen, which contaminated the polymers, are presented in Fig.3.

The asymmetric spectrum predominant in PTTF A.2 and A3 assigned to immobile radicals R', has a typical powder pattern caused by an anisotropic g-tensor with a symmetry lower than axial. The mean g-factor is g = 2.0081. Simulation of the measured spectra of Fig. 3 by superposition of an asymmetric spectrum with $g_x = 2.0147$, $g_z = 2.0067$ and $g_v = 2.0028$ and a symmetric one of mobile radicals $R^m = R_2^m$ with g = 2.0071 yields that the paramagnetic centres of the undoped samples consist at least to 10 % of the radicals R^i . In the samples A.2 and A.3 the portion of R^i radicals even amounts to 54 % and 75 %, respectively.

148



Fig 3: 3 cm wave band ESR spectra of two "undoped" PTTF samples in which immobile radicals R' predominate, measured at room temperature.

The mobile cation radicals R_2^{m} (or polarons P⁺*) have a *g*-factor between g = 2.0068 (B.I) and g = 2.0078 (A.I) depending on the molecular structure of the PTTF (Ref.15). These values are in agreement with the *g*-factors of cation radicals in the tetrathiaflilvalene (TTF) (Ref.19). The PTTF A.1 and B.I have a nearly plane molecule structure. Due to this structure an electronic interaction occurs between the TTF units in the polymer, so that in an oxygen contaminated planar PTTF an electron transfer from the PTTF system to the ³O₂ occurs, predominantly forming a TTF radical cation R_2 , which has the same structure as a radical cation formed by doping with iodine (Ref.15). The structure of R^{i} can be realized by a homopolar bond between the oxygen and a sulfur atom of the TTF ring (forming a radical R_1) or by an ion pair interaction between the cation radical R_2 and the counterion O_2^{-} forming R_2^{T} , i.e., a trapped cation radical. A possible structure of R_1 , and that of R_2 can be represented by the following formulas (Refs. 16,17).



Because sulfur has a relatively large spin orbit coupling with $1 = 382 \text{ cm}^{-1}$, an asymmetric spectrum is to be expected in trapped cation radicals R_2^T . Nevertheless, it cannot be excluded that the observed asymmetric spectrum is partly due to radicals of type R_1 (Ref. 16). The addition of iodine as a dopant causes a drastic change of the line shape in outcome of a radical conversion. At very low dopant concentration the iodine atoms effect mainly a conversion of the localized radicals into mobile cation radicals. This results in the increase of conductivity without increasing the total spin concentration. The mobile radicals R_2^m directly contribute to the conductivity as polarons or indirectly by recombination to bipolarons, thus losing their spin paramagnetism. In the case of low dopant concentration the line shape of the ESR spectra shows that there are still localized paramagnetic centres in the samples. The simulation of ESR spectra measured at room temperature has shown that the addition of 0.2 iodine atoms per monomeric unit effects that in the time scale of 3 cm ESR the spectrum is to more than 90 % due to mobile R_2 radicals and to less than 10 % due to localized radicals.

3.2. Results of ESE measurements

In the used equipment (Ref.20) only those paramagnetic centres contribute to the ESE detected ESR spectrum which has a phase memory time $T_{\rm M} > 250$ ns. From the radicals in PTTF discussed up to now, only the localized or trapped radicals R_1 and $R_2^{\rm T}$ were detected. The ESE detected ESR spectra of the phenylene-bridged PTTF A3 are presented in Fig.4 (Ref. 16).



Fig.4: ESE detected ESR spectra of PTTF A.3 recorded in 3 cm wave band at 110 K and 140 K from an "undoped" sample () and two iodine doped samples containing 0.2 and 0.4 iodine atoms per monomeric unit which corresponds to y = 0.06 and 0.12, respectively.

In the case of the not iodine doped or "undoped" sample the magnetic field distances between the singular points of this powder pattern agree well with those of the ESR spectrum of the immobile radicals R^i trapped by reaction or interaction with oxygen $(R_1, R_2^T/O_2^-)$ which, predominantly determines the shape of the spectra A. 2 and A.3 in Fig.3 of the "undoped" PTTF. However, there are significant differences in the g-value anisotropy detectable by ESE of the immobile radicals in the weakly doped samples of A.3 if the dopants are oxygen and iodine, respectively. Probably the anisotropic spectrum in iodine doped samples is due to localized radical R_2^T trapped by an interaction with the counter anion $I_x^ (R_2^T/I_x^-)$. Also from the chemical viewpoint it can be assumed the peroxy species cannot be present in the iodine doped samples because I_x^- is a reducing anion and would destroy R_1 , R_2^T/O_2^- (Refs. 16,17). The number of immobile radicals N_R decreases with rising temperature T according to

$$N_{\rm R} = N_{\rm R}^0 \exp[-E_{\rm A}/(kT)]$$

with *k* - Bollzmann constant, and an activation energy $E_A = 0.01$ eV, equal for both samples weakly doped with iodine.

The basic idea of interpretation is that with increasing temperature the $R_2^{\rm T}/I_x^{-}$ lose their trapping state and become mobile cation radicals $R_2^{\rm m}$ (or polarons) with such a short relaxation time that they are undetectable by the ESE spectrometer applied or they form bipolarons thus losing their spin paramagnetism. The activation energy E_A for the decrease of localized radicals is of the same order of magnitude as the E_A of increasing electrical conductivity in this temperature region ($E_A = 0.03 \text{ eV}$). This points out that the radical conversion mentioned really contributes to the increase of conductivity measured in the low temperature range (Ref. 16).

3.3. Structure and temperature dependent polaron/bipolaron conversion

Both ¹²⁷I-Mössbauer and Raman spectroscopies find that iodine intercalates in poly(tetrathiafulvalenes) in the form of polyiodine species, namely I_3^- and I_5^- . The relative I_3^- content decreased with rising doping level. Below 0.5 iodine per monomeric unit, iodine is present mainly as I_3^- molecules (Ref. 21). The I_3^- (and I_5^-) anions are found to be more strongly bound in the PTTF polymers than in polyacetylene. This behaviour is also reflected by their Raman frequencies. We can estimate the charge carrier density from the dopant concentration. If we assume the predominant formation of I_3^- as counterions, every third iodine atom takes an electron from the PTTF generating a cation; in the samples there is a minimum of about 10^{21} charge carriers per gram. The radical concentration amounts to max. 10^{19} spins per gram. That means the spin density in the doped PTTF is almost two orders

of magnitude lower than the charge density.

From these ESR measurements it follows that there are on average less mobile radicals $R_2 = P^{+\bullet}$ in the sample than macromolecules. If a second polaron $P^{+\bullet}$ is formed on a macromolecule two polarons can form a diamagnetic bipolaron B^{2+}

$$P^{+\bullet} + P^{+\bullet} \rightarrow B^{2+}$$

This process takes place with a higher probability in molecules with plane structure where electron interaction is favoured between the monomeric units. This can be assumed because the samples A.1 and B. 1 with almost plane structure show a lower radical concentration than the polymers A.2 and A.3 (which are of a twisted molecular structure), but the samples A.1 and B.I have the highest conductivity. These results support the assumption that the bipoiarons generated via polarons provide the main contribution to the conductivity of the doped PTTF (Refs.16,17). The following formulas represent possible structures of polarons and bipolarons in the PTTF samples B. 1.



The decrease in temperature from room temperature to 150 K leads to a slight decrease in the signal amplitude supposed because of the bipolaron formation. However, below 150 K an increase in the signal was observed in some PTTF. The observed variation of radical concentration with decreasing temperature leads us to the assumption that there is a temperature-dependent equilibrium between polarons and bipolarons which tends to shift again more to the polaron-side in the low-temperature region below 150 K (Ref.16).

4. Studies on PTTF by high-field ESR in the 2 mm wave band

In Ref.14 both high spectral resolution and saturation effects in the 2 mm wave band ESR have been used for a more correct determination of the nature and dynamics of trapped and mobile paramagnetic centres (PC) in one slightly doped PTTF sample. With this investigation we demonstrated also the possibility of studying superslow anisotropic molecular reorientations in conjugated polymers using the 2 mm wave band saturation transfer ESR (ST-ESR) method (Ref.9).

4.1. Determination of ESR parameters

ESR experiments were performed using a 3 cm (PS 100.X) wave band ESR spectrometer and a 2 mm (ESR 5-01) spectrometer with a superconductive magnet described in detail in Ref.22. The powder-like PTTF A.2 sample (with a doping level y = 0.12 (or 0.4:1), total spin concentration of $1-10^{18}$ spin/cm³ and d.c. conductivity of 10^5 S cm⁻¹) has been investigated (Ref.14).

In the 3 cm wave band the PTTF sample shows a nearly symmetrical signal which seems to be not influenced by immobilized PCs (Fig. 5a). Individual components of immobilized PCs are manifest more evidently in the 2 mm ESR spectrum. In fact, the 2 mm wave band ESR absorption spectra of the sample are more informative (Fig. 5b). They allow a correct determination of all major components of the anisotropic g-tensor and the separation of the lines of different shapes.



Fig.5: ESR spectra of the PTTF sample A.3 doped with 0.4 iodine atoms per monomeric unit measured at 110 K; a) in the 3 cm wave band, b) in the 2 mm wave band.

Computer simulation shows that the signal is a superposition of a strongly asymmetric spectrum with temperature independent magnetic parameters of $g_x = 2.01424$, $g_y = 2.00651$, $g_z = 2.00235$ and a symmetric one with $g_P = 2.00706$. As measured in the time scale of 140

GHz or 2 mm ESR at low temperatures the two types of PCs are present in the sample with a concentration ratio of 1:1.8, or $n_1 = 3.8 \times 10^{-5}$ and $n_2 = 6.9 \times 10^{-5}$ spin per monomeric unit, respectively.

The *g*-factor g_p of the symmetric part of the spectrum is close to the average *g*-factor of the immobile PCs determined as $\langle g \rangle = 1/3 (g_x + g_y + g_z) = 2.00770$. It is known (Refs.7,9,23) that the principal components of the *g*-tensor of PCs in some conjugated polymers could be averaged if the PCs are moving with a rate $v_{1D}^0 \ge (g_x - g_z)\beta B_0/h$.

In PTTF there exist two types of PCs with approximately equal magnetic parameters, they are polarons moving along the polymer chain (X-axis) with the rate of $v_{1D}^0 \ge 5 \cdot 10^9 \text{ s}^{-1}$ and polarons pinned on short polymer chain segments. The part of mobile polarons increases with rising dopant concentration (Refs. 16,17).

The measurements of Krinichnyi et al. (Ref.14) have shown that the linewidth of ESR spectral components of immobilized polarons, ΔB_{pp} , is temperature-independent and may vary within 0.28 - 0.38 and then to 3.9 mT while the operation microwave frequency n_e increases from 9.5 to 37 and then to 140 GHz, respectively. The line width of mobile polarons, ΔB_{pp} , in this case increases at 270 K from 1.0 to 1.12 and then to 17.5 mT, respectively. The line width of both types of PCs in PTTF increases quadratically only in the 37 GHz $\leq v_e \leq 140$ GHz frequency region. This fact is discussed in terms of the non-interacting spin-packets in Ref.14. In this paper the fact that the mobile PCs have a broader line than the immobile ones is explained by the strong interaction with dopants. So the mobile PCs have probably a stronger interaction with the dopants than the immobile PCs. This is another interpretation than given in Refs. 16,17.

The decrease in the spectrum line width of the mobile polarons by 2.2 mT together with the 2fold temperature decrease indicates that the ID-mobility of polarons becomes more intensive as the temperature falls and the so-called motional narrowing of the line takes place. Such a change in ΔB_{pp} is very similar to the temperature dependence of the line width in metals.

4.2. Spin-dynamics in PTTF

Spin-packets in paramagnetic systems are characterized by spin-lattice and spin-spin relaxation times T_1 and T_2 , resonance frequencies ω_i and exchange frequency v_{ex} . In a reasonably strong magnetic field the inequality $v_{ex} < \Delta \omega_{ij}$ can be satisfied, so that the line broadens inhomogeneously with the total width of $\Delta B_{PP} = 1/(\gamma_e T_2) + \Delta \omega_{ii}^2/8\gamma_e v_{ex}$ (Ref.24) and is described by the Gaussian distribution function of separate non-interacting spin-packets. The passage effects are observed in the ESR spectrum under the condition of microwave saturation of spin-packets:

$$\gamma_{\rm e} B_1 \sqrt{T_1 T_2} > 1 \tag{1}$$

and the well-known conditions of adiabatic resonance passage

$$dB/dt = B_{\rm m}\omega_{\rm m} < \gamma_{\rm e}B_{\rm l}^2 \tag{2a}$$

$$\gamma_{\rm e} B_{\rm m} << \Delta \omega_{\rm G} \tag{2b}$$

$$\omega_{\rm m} T_1 \le 1 \tag{2c}$$

where γ_e is the gyromagnetic ratio for electron, dB/dt is the rate of resonance passage, B_m and ω_m are the a.c. modulation amplitude and angular frequency, respectively, and $\Delta\omega_G$ is the width of the Gaussian function of spin-packet distribution. In this case the precession period of the magnetization vector \boldsymbol{M} of PCs around the vector B_1 becomes smaller than the effective electron relaxation time $T_{eff} = \sqrt{T_1T_2}$ as a result the relaxation processes within the resonance conditions do not produce a strong effect on the behaviour of \boldsymbol{M} rotation.

We have measured the relaxation times T_1 and T_2 of both types of PCs in the PTTF sample as function of temperature, determined from the 2 mm ESR spectra registered under the conditions (1) and (2). Thus T_1 proves to change with temperature like T^{α} , where $\alpha = 3.3$ and 3.7 for localized and mobile polarons, respectively. The exponent a determined from the 2 mm wave band ESR is larger than that measured for immobile radicals using the 3 cm ESE technique (Ref. 16,17). This deviation may be explained by the circumstance, that in the 2 mm wave band many radicals appear immobile PCs which in the 3 cm wave band are counted to the mobile ones (because of the lower measuring frequency). For that reason also the absolute values of relaxation times are shorter in the 2 mm wave band compared with that measured for the immobile radicals in the 3 cm wave band (Ref. 16). The small difference between T_1^{mob} and T_1^{imm} may be a hint of a strong interaction and energy exchange between mobile and immobile PCs (Ref. 14).

The motion of chains in conjugated polymers, similar to the ordinary ones, is *a priori* strongly anisotropic with the correlation time $\tau_c \ge 10^{-7}$ s. According to the saturation transfer (ST) ESR approach, if conditions (1) and (2) hold, the chain reorientation causes that spins pass through the resonance condition because of the anisotropy of the magnetic interaction. If a slow molecular motion around, e.g., the X-axis occurs, the inequality (1) may be not satisfied for radicals oriented by their Y or Z axis parallel to the direction of the external magnetic field (Ref. 14).

As shown in Ref. 12 the sensitivity of the ST-ESR method to a very slow molecular motion increased at the 2 mm wave band, and allows to study the relaxation and anisotropic motion

155

processes separately. In this case the PCs trapped in PTTF are sensors of a polymer chain segment motion. Because the sensitivity of the ST-ESR method to molecular motion is enhanced quadratically with the magnitude of the anisotropy of the magnetic interaction, the 2 mm wave band ST ESR method is particularly efficient for the investigation of PTTF by means of the immobilized PCs with a strongly anisotropic g-tensor. In Ref. 14 is reported on the detection of a very slow anisotropic librational reorientation of the immobilized PCs together with the polymer chains. The correlation time of such libration is temperature-dependent according to $\tau_c = 9.8 \cdot 10^{-6} \exp[-0.02 \text{ eV}/(kT)]$.

4.3. Diffusive intrachain polaron motion and interchain polaron hopping, on-chain and interchain conductivities

It should be noted that the activation energy for such librations is comparable with that of d.c. conductivity in the low temperature region of a weakly doped PTTF sample (Ref.15), which indicates the interaction of immobilized and mobile polarons in PTTF. In comparison of our experimental results with the polaron theory, we assume the on-chain quasi-one-dimensional (ID) diffusive motion of mobile PCs in conjugated chain segments of PTTF with the rate of v_{1D} and their expansion by three-dimensional (3D) hopping between the chains with the frequency of v_{1D} .

The temperature dependences of v_{1D} and v_{3D} determined in Ref.14 are plotted in Fig.6. The extrapolation to room temperature displays that the maximum value of v_{1D} of the PTTF sample does not exceed 5-10⁹ s⁻¹. Taking into account the fact that the spin delocalization of a polaron in conjugated polymers occupies approximately three to five monomeric units (Ref.25), then $v_{1D} \leq 2 \cdot 10^{10} \text{ s}^{-1}$ would be correct. This value is lower than that determined for polarons in polypyrrole (Ref.26) and polyaniline (Refs.1 1,27).

The spin dynamic behaviour can be related to the charge transport properties. At the equal diffusion coefficients for both spin and charge carriers, the PTTFs on-chain and interchain conductivities can be estimated from v_{1D} and v_{3D} using the Einstein relation

$$\sigma_{1D,3D} = n_2 e^2 v_{1D,3D} d^2 / (kT)$$
(3)

where *d* is the lattice constant and *k* is the Boltzmann constant. With $n_2 = 6.9 \cdot 10^{-5}$, one obtains the values of $\sigma_{1D} \approx 1 \times 10^{-3}$ and $\sigma_{3D} \approx 4 \times 10^{-5}$ S cm⁻¹ for the intrinsic on- and interchain conductivities, respectively, in the room temperature region (Ref.14). The behaviour of $\sigma_{1D}(T)$ (and of $\Delta B_{pp}^{mob}(T)$) dependences suggest a metallic state for PTTF, but this is only valid for the intrachain conductivity. Only the σ_{3D} -value is in agreement with $\sigma_{dc} \approx 10^{-5}$ S cm⁻¹ as measured by electrical methods for PTTF. That means σ_{dc} is determined by interchain spin hopping.



Fig.6: Frequencies v_{1D} and v_{3D} of diffusive on-chain polaron motion and of interchain polaron hopping, respectively. Quasi-1D-intrachain conductivity σ_{ID} and 3D-mterchain conductivity σ_{3D} as estimated by ESR.

Because the E_A value of the dc-conductivity in the low temperature region corresponds to the above determined activation energy for PTTF chains libration, $E_a = 0.02 \text{ eV}$, we conclude that the conductivity of PTTF is determined mainly by a phonon-assisted interchain hopping of polarons. It is stimulated by multiphonon processes and it is supported by the libration of polymer chains due to the strong electron-lattice coupling (Ref.14).

5. Conclusions

The discussed investigation examples have shown that ESR spectroscopy is a valuable tool for getting deeper insight in the interesting electronic and magnetic properties of conducting polymers.

The study of spin dynamics by ESR in the 2 mm wave band allows a separate investigation of the quasi-1D-intrachain polaron diffusion and of the 3D-interchain polaron hopping. In PTTF, as in many other semiconducting polymers, a large anisotropy of the electron or polaron

motion is to be observed in the whole temperature region and at all doping levels, whereas in highly doped PANI, near room temperature, the anisotropy of polaron motion disappears. That means $v_{1D} \approx v_{3D}$ which supports the assumption of metallic islands in PANI.

References

- H.-K. Roth, F. Keller, H. Schneider: "Hochfrequenzspektroskopie in der Polymerenforschung", Akademie-Verlag Berlin 1984
- (2) L.V. Hinh, G. Schukat, E. Fanghanel, <u>JPrakt.Chem.321</u>.299(1979)
- (3) V.Q. Trinh, L.V. Hinh, G. Schukat, E. Fanghanel, J.Prakt Chem. 331. 826 (1989)
- F. Lux, Ph.D. Thesis, TU Berlin, 1992
 F. Lux, G. Hinrichsen, M.-H. Pohl, proceedings of the RABRA-meeting on conducting polymers, Bristol 1992
- (5) Y. Cao, A. Andretta, A.I. Heeger, P. Smith, <u>Polymer</u> 30,2305(1989)
- (6) K.L. Tan, B.T.G. Tan, S.H. Khor, KG. Neoh, E.T. Kang, <u>J.Phys.Chem.Solids</u> 52, 673 (1991)
- L.M. Goldenberg, A.E. Pelekh, V.I. Krinichnyi, O.S.Roschupkina, A.F. Zueva, R.N. Lyubovskaya, O.N. Efimov, <u>Svnth.Met. 36</u>.217 (1990)
- (8) A.E. Pelekh, L.M. Goldenberg, V.I. Krinichnyi, <u>SvnthMet. 44</u>.205 (1991)
- A.E. Pelekh, V.I. Krinichnyi, A.Yu. Brezgunov, L.I. Tkachenko, G.I. Kozub, <u>Mater.Sci.17</u>. 25 (1991). <u>Vvsokomolek. Soedin.33.</u>1731 [[991])
- (10) V.I. Krinichnyi, A.E. Pelekh, L.I. Tkachenko, G.I. Kozub, <u>Svnth.Met.46</u>.1 (1991)
- (11) V.I. Krinichnyi, I.B. Nazarova, S.D. Chermisow, F. Lux, G. Hinrichsen, H.-K. Roth, publication in preparation
- (12) V.I. Krinichnyi, J.Biochem./Biophys.Methods 23.1 (1991)
- (13) V.I. Krinichnyi, A.E.Pelekh, L.I.Tkachenko, G.I. Kozub, <u>Svnth.Met. 46</u>.13 (1992)
- (14) V.I. Krinichnyi, A.E.Pelekh, H.-K. Roth, K. Lttders, ADDI. Maen. Res, in press
- (15) H.-K. Roth, H. Gruber, E. Fanghunel, Trinh vu Quang <u>Progr.Colloid.Polvm.Sci.78.75</u> (1988)
- (16) H.-K. Roth, H. Gruber, G. VOlkel, W. Brunner, E. Fanghanel, <u>Progr.Colloid.Polvm</u>, <u>Sci. 80</u> 254 (1989)
- (17) H.-K. Roth, W. Brunner, G. Volkel, M. Schrddner, H. Gruber, <u>Makromol.Chem..Macromol.Svmp. 34.293</u> (1990)
- (18) F. Moreas, D. Davidov, M. Kobayashi, T.C. Chung, J. ChengA-J Heeger, F. Wudl, <u>Svnth. Metals 10</u>,169 (1985)
- (19) H.G. Fitzky, J. Kocker, <u>Svnth.Metals 13.</u>335 (1986)
- (20) W. Brunner, F. Genttzsch, Exper. Techn. Phys. 19.39 (1971)